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The carbon chemical shift tensor in polyoxymethylene

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The complete chemical shift tensor of the carbons in crystalline polyoxymethylene is determined via the two-dimensional dipolar correlated $^{13}$C chemical shift NMR technique. The orientation of the tensor is determined by the local tetrahedral symmetry of the $C_2H_2O_2$ unit and not by the overall symmetry of the helical chains. In an accompanying paper this shift tensor is used to study ultraslow motions in polyoxymethylene.

I. INTRODUCTION

Many spin interactions in a solid are described by a tensor. As tensor interactions are related to molecular and crystal axes, the information contained in such a tensor interaction can be used to obtain geometrical information about the molecule or crystal being studied. It is well known, e.g., that the magnetic dipole interaction between two spins is coaxial with the internuclear axis of the two interacting nuclei. Knowledge of the complete tensor of the dipolar interaction between a $^{13}$C and a $^1$H, for instance, yields quantitative information about the C–H distance and the orientation of the C–H axis, with respect to the molecule.

The first technique to record dipolar modulated $^{13}$C spectra of solids, the 2D separated local field (SLF) experiment, was proposed by Waugh and applied to single crystals and oriented polymers by his co-workers. Here, dipolar modulation refers to the fact that $^{13}$C FIDs are collected during the detection period of the 2D experiment, which are phase modulated by the dipolar interaction of the $^{13}$C spin with surrounding protons. It is well known, e.g., that the presence of anisotropic chemical shifts in a solid state NMR experiment gives rise to a signal modulation. The orientation of this modulation refers to the fact that the $^{13}$C spin orientation relative to the molecular axes yields a signal which is also modulated by the dipolar interaction of the $^{13}$C spin with surrounding protons. Since the orientation of the latter is known, the $^{13}$C chemical shift tensor can be determined relative to molecular axes. Here we apply this technique to polyoxymethylene, in order to be able to analyze slow molecular motions in this polymer, as described in the next paper.

II. EXPERIMENTAL

The pulse scheme used to obtain spectra which correlate heteronuclear dipolar interactions and chemical shielding anisotropy is shown in Fig. 1. Multiple pulse decoupling (MREV-8, $\pi/2$ pulse length 1.6 $\mu$s, cycle time 28.8 $\mu$s) was used for homonuclear proton decoupling during $t_1$. The scaled heteronuclear $^{13}$C–$^1$H dipolar interaction and the anisotropic chemical shift modulate (in phase) the signal detected during $t_2$. As proton decoupling is applied during $t_2$ only the anisotropic chemical shift is effective during this detection period.

For a semicrystalline polymer like polyoxymethylene the presence of the NMR signal from amorphous parts of the material interferes with the dipolar correlation spectrum of the crystalline regions. The amorphous signal is eliminated by the pulse sequence of Fig. 1 by making use of the vastly different spin–lattice relaxation times $T_1$ for the amorphous and crystalline carbons. At 75 MHz, the carbon frequency of the CXP 300 spectrometer used for this experiment, the amorphous carbon $T_1$ is 90 ms, while the crystalline $T_1$ is 18 s. After cross polarization the total magnetization (amorphous + crystalline) is flipped to the $-z$ axis from where the amorphous part of the magnetization rapidly relaxes to the $+z$ axis. At the point where this magnetization passes through zero another $\pi/2$ pulse is applied which brings the crystalline magnetization in the $xy$ plane from where the evolution period for the 2D experiment starts.

The double-resonance probe was equipped with a horti-
The 9/5 model, meaning that the unit cell consists of chains with C-H plane makes an angle of 55°. It is difficult to grow large crystals of the polymer, but some x-ray form of studies of small single crystals have been made. The stable amorphous material. The lamellae, in which the molecular crystallinity is approximately 100%, have trigonal symmetry, with the chains arranged in a helical conformation. Within a single crystal only helices with the same handedness are found, with right or left handedness equally probable. Huggins proposed the 9/5 model, meaning that the unit cell consists of chains with nine CH₂-O monomer units in five turns of the helix. Uchida and Tadokoro refined this to a 29/16 model. Both helices are so similar, however, that it is difficult to decide which model is the most reasonable. The local tetrahedral geometry of the monomer units in five turns of the helix. The dipolar correlated chemical shift spectrum for the crystalline parts of polyoxymethylene, free from the signal of the polymerization product of formaldehyde.

### III. The structure of polyoxymethylene

Polyoxymethylene, with chain structure (-CH₂-O-)ₜ, is the polymerization product of formaldehyde. It is rather difficult to grow large crystals of the polymer, but some x-ray studies of small single crystals have been made. The stable form of POM has trigonal symmetry, with the chains arranged in a helical conformation. Within a single crystal only helices with the same handedness are found, with right or left handedness equally probable. Huggins proposed the 9/5 model, meaning that the unit cell consists of chains with nine CH₂-O monomer units in five turns of the helix. Uchida and Tadokoro refined this to a 29/16 model. Both helices are so similar, however, that it is difficult to decide which model is the most reasonable. The local tetrahedral geometry of the CH₂O unit appears to be hardly distorted by the helix formation (O-C-O bond angle ~ 111°). The H-C-H angle is 109° and the C-H bond length is 1.09 Å. The H-C-H plane makes an angle of 55° with the helix axis. This information about the proton positions allows us to find the orientation of the chemical shift tensor within the molecule.

Bulk-crystallized POM has a spherulitic structure, consisting of crystalline lamellae connected to each other by amorphous material. The lamellae, in which the molecular chains are folded, are approximately 100 Å thick. The crystallinity is approximately 60%. A detailed description of a spherulitic structure in crystalline polymers is given by Sharples.

Proton-decoupled ¹³C spectra of a static sample of bulk-crystallized POM at room temperature show separate signals of the crystalline and the amorphous phase. Because the glass temperature of the amorphous phase is below room temperature, a lot of motion exists in this part of the polymer. In fact, there is so much motion in the amorphous material that the chemical shift anisotropy is largely averaged, which results in a fairly narrow line in the NMR spectrum. For the rigid crystalline regions in the sample we observe a structured line determined by the chemical shielding anisotropy. From this line shape the principal values of the chemical shift tensor were found to be: \( \sigma_{xx} = 67 \) ppm, \( \sigma_{yy} = 86 \) ppm, and \( \sigma_{zz} = 111 \) ppm.

### IV. Experimental result and simulation of the dipolar chemical shift correlated spectrum

The dipolar correlated chemical shift spectrum for the crystalline regions of polyoxymethylene, as obtained with the pulse program of Fig. 1, is shown in Fig. 2(a). We have to find the relative orientation of the carbon chemical shift tensor by simulation of this spectrum for different assumed relative orientations of the ¹³C-¹H dipolar and ¹³C chemical shift tensors.

A general description of the calculation of powder resulting from experiments correlating tensorial interactions is given by Linder et al. Here we describe the specific case of the correlation of the heteronuclear dipolar interaction to the chemical shielding anisotropy in a CH₂ group of polyoxymethylene. During the evolution period \( t_e \), the strong homonuclear ¹H-¹H is suppressed by multiple pulse decoupling. Thus, the relevant Hamiltonian for the ¹³C chemical shift is determined by the chemical shift \( H_{CS} \) and the heteronuclear dipolar interaction \( H_D \). During the detection period normal high-power proton decoupling is applied so that only the ¹³C Zeeman interaction remains:

\[
H^{(1)} = H_{CS} + CH_D,
\]

\[
H^{(2)} = H_{CS},
\]

where \( C \) represents the scaling factor due to the homonuclear decoupling. Of course it is possible to remove the influence of \( H_{CS} \) during \( t_e \) by applying a \( \pi \) pulse in the middle of the evolution period, as has been described by Stoll et al. But the effect of \( H_{CS} \) during \( t_e \) can also be calculated, therewith avoiding the need of an accurately adjusted \( \pi \) pulse in the experiment.

Evaluation of the Hamiltonians \( H^{(1)} \) and \( H^{(2)} \) gives us the resonance frequencies during evolution and detection period, thus allowing the calculation of the total 2D powder spectrum.

**FIG. 1.** Pulse scheme to obtain a 2D dipolar chemical shift correlated spectrum of the crystalline parts of polyoxymethylene, free from the signal of the amorphous regions.

**FIG. 2.** (a) 2D dipolar correlated ¹³C spectrum of crystalline polyoxymethylene obtained with the pulse scheme of Fig. 1. MREV-8 (\( \pi/2 \) pulse length 1.6 \( \mu s \), cycle time 28.8 \( \mu s \)) was used during \( t_e \). 400 scans were accumulated for 34 \( t_e \) values. In \( t_e \), a cosine filter was applied prior to Fourier transformation. (b) Simulation of spectrum (a) with the orientation of the chemical shift tensor in POM (see the text). \( \alpha = 180°, \beta = 90°, \gamma = 90°. \)
pattern. Summing over all transitions \((i,j)\), with resonance frequencies \(\omega_{ij}^{(1)}(\theta,\varphi)\) and \(\omega_{ij}^{(2)}(\theta,\varphi)\), we get

\[
S(\omega_1,\omega_2) = \sum_i \sum_j \int_0^{2\pi} \int_0^\pi g_{ij}^{(1)}[\omega_1 - \omega_{ij}^{(1)}(\theta,\varphi)]g_{ij}^{(2)}[\omega_2 - \omega_{ij}^{(2)}(\theta,\varphi)] \sin \theta d\theta d\varphi,
\]

where \(\theta\) and \(\varphi\) are the polar angles that describe the orientation of randomly distributed microworlds with respect to the external magnetic field. The line shapes are denoted by the functions \(g\).

The resonance frequency of a certain \(^{13}\)C spin, relative to the frequency \(\gamma B_0\) during the evolution period depends on the eigenstates of the two connected protons and on the orientations of the chemical shielding and dipolar tensors with respect to the external field:

\[
\begin{align*}
\omega_1^{(1)} &= \omega_{CS} + \omega_{D_1}(\gamma,\varphi,\alpha,\beta,\gamma) \\
\omega_1^{(2)} &= \omega_{CS} - \omega_{D_2}(\gamma,\varphi,\alpha,\beta,\gamma) \\
\omega_2^{(1)} &= \omega_{CS} + \omega_{D_1}(\gamma,\varphi,\alpha,\beta,\gamma) \\
\omega_2^{(2)} &= \omega_{CS} - \omega_{D_2}(\gamma,\varphi,\alpha,\beta,\gamma).
\end{align*}
\]

During the detection period we simply find

\[
\omega_{ij}^{(2)} = \omega_{CS}(\theta,\varphi).
\]

Here \(\theta\) and \(\varphi\) are the polar angles which orient the external magnetic field in the principal axis system of the chemical shielding tensor. \(\alpha, \beta, \gamma\) are the Euler angles relating the dipolar interaction to the principal axis system of the CS tensor. The expression for \(\omega_{CS}\) is easily obtained by transforming the Hamiltonian \(H_{CS}\) from the principal axis system to the laboratory frame over the Euler angles \(\varphi, \theta, 0:\)

\[
\omega_{CS}(\theta,\varphi) = \omega_{xx} \sin^2 \theta \cos^2 \varphi + \omega_{yy} \sin^2 \theta \sin^2 \varphi + \omega_{zz} \cos^2 \theta,
\]

with \(\omega_{xx} = \gamma B_0 \delta_{xx}\), etc. In order to obtain expressions for \(\omega_{D_1}\) and \(\omega_{D_2}\) for two protons, we have to evaluate the dipolar Hamiltonian \(H_D\) in the laboratory frame. To get similar expressions for both protons we define a local axis system \((LOC)\) in the \(\text{CH}_2\text{O}_2\) unit whose \(z\) axis is the bisector of the \(\text{H}--\text{C}--\text{H}\) angle and the \(x\) axis is perpendicular to the \(\text{H}--\text{C}--\text{H}\) plane (Fig. 3). To come from the principal axis system of the dipolar interaction \((D-PAS)\) to this axis system, one has to rotate over an angle \(\delta\) about the \(y\) axis for proton 1 and \(-\delta\) for proton 2 \((\delta = \text{H}--\text{C}--\text{H} \text{ angle} = 109.5^\circ)\). As we like to determine the orientation of the chemical shielding tensor within the molecule, we transform from this local axis system to the principal axis system of the chemical shielding tensor \((CS-PAS)\) over the Euler angles \(\alpha, \beta, \gamma\). These angles \(\alpha, \beta, \gamma\) are thus the unknown parameters which we want to recover from the simulated spectra. The last step is then to transform from the CS-PAS to the laboratory frame \((LAB)\) over angles \(\varphi, \theta, 0:\)

\[
D-PAS \rightarrow \text{LOC} \rightarrow \text{CP-PAS} \rightarrow \text{LAB},
\]

\[
H_D = \sum_m D_{m0}^2 \gamma_0 \sum_m D_{m,m}^2 (\alpha, \beta, \gamma) \times D_{m,m}^2 (0, \pm \delta/2, 0) \gamma_c \gamma_H \hbar^2 r_{CH}^2 I_S I_x.
\]

Some algebra eventually yields the contributions of the dipolar interactions to the resonance frequencies during \(t_1:\)

\[
\omega_{D,1,2}^{(1,2)} = -\frac{1}{2}\gamma_c \gamma_H \hbar \times (1 - 3[A \sin (\pm \frac{1}{2}\delta) + B \cos (\pm \frac{1}{2}\delta)]^2),
\]

with

\[
A = \sin \theta \cos \alpha \beta \cos (\varphi + \gamma) - \sin \alpha \sin (\varphi + \gamma),
\]

\[
B = -\sin \theta \beta \cos (\varphi + \gamma) + \cos \theta \cos \beta.
\]

Before we can calculate the 2D powder pattern from these resonance frequencies using Eq. (2), there is still one point to consider, namely the line shape for each resonance. This depends on the type of modulation and how the data are Fourier transformed. In this case we have to do with phase modulation, i.e., the phase of the signal detected during \(t_1\) is a linear function of length of evolution period \(t_1\). Successive complex Fourier transforms with respect to \(t_2\) and \(t_1\) gives the following line shape of mixed absorption and dispersion signals:

\[
S(\omega_1,\omega_2) - \{A(\omega_1)A(\omega_2) - D(\omega_1)D(\omega_2) + i[A(\omega_1)D(\omega_2) + D(\omega_1)A(\omega_2)]\}
\]

with

\[
A(\omega_i) = \frac{T_z}{1 + T_z^2 (\omega_i - \omega)^2}
\]

and

\[
D(\omega_i) = \frac{T_z^2 (\omega_i - \omega)}{1 + T_z^2 (\omega_i - \omega)^2}.
\]

Generally an absolute value calculation is performed on such a signal, i.e., the square root of the sum of the squares of the real and the imaginary signal is calculated.

The presence of dispersion mode signals lowers the resolution of a powder spectrum, and can even lead to the disappearance of certain ridges in a 2D powder pattern. Pure absorption mode spectra can be achieved by performing a second experiment for each \(t_1\) value with an additional \(^{13}\)C \(\pi\) pulse at the end of the evolution period and coaddition of the thus obtained FID to the FID of the original experiment. The limited number of lines in a pulse program on a Bruker

---

**FIG. 3.** Local axis system LOC in the CH2O2 unit. The O--C--O plane is chosen as the \(xz\) plane, and the H--C--H plane as the \(xy\) plane. The helix axis (---) makes an angle of 55° with the H--C--H plane.
FIG. 4. The simulated spectra for the six possible orientations of the shift tensor for which one principal axis is parallel to the helix axis and another in the O–C–O plane. (a) $\alpha = 55.2^\circ$, $\beta = 0^\circ$, $\gamma = 0^\circ$, $\sigma_{22}$//helix, $\sigma_{33}$//OCO; (b) $\alpha = 55.2^\circ$, $\beta = 90^\circ$, $\gamma = 0^\circ$, $\sigma_{22}$//helix, $\sigma_{11}$//OCO; (c) $\alpha = 55.2^\circ$, $\beta = 90^\circ$, $\gamma = 90^\circ$, $\sigma_{11}$//helix, $\sigma_{22}$//OCO; (d) $\alpha = -34.8^\circ$, $\beta = 0^\circ$, $\gamma = 0^\circ$, $\sigma_{11}$//helix, $\sigma_{33}$//OCO; (e) $\alpha = -34.8^\circ$, $\beta = 90^\circ$, $\gamma = 90^\circ$, $\sigma_{33}$//helix, $\sigma_{11}$//OCO; (f) $\alpha = -34.8^\circ$, $\beta = 90^\circ$, $\gamma = 90^\circ$, $\sigma_{33}$//helix, $\sigma_{22}$//OCO.

The 2D spectra were calculated in a rectangular grid of $100 \times 100$ frequency values $(\omega_1, \omega_2)$. Generally 90,000 sampling points were evaluated over a whole sphere. The spectrum thus calculated is subsequently convoluted with the line shape function described in Eq. (8) followed by an absolute value calculation, resulting in a spectrum that can be compared to the experimental one.

From the structure of polyoxymethylene one expects the orientation of the chemical shielding tensor to be determined either by the local tetrahedral symmetry of the CH$_2$O unit or by the overall symmetry of the helix (or even by both). Simulation of these possibilities gives very different 2D powder patterns, and the right solution is easily determined. Figure 4 shows the calculated spectrum for the six possible orientations of the carbon tensor for which one principal axis is parallel to the helix axis and another in the O–C–O plane. It is clear that none of these simulated spectra even resembles the experimental spectrum of Fig. 2(a).

Figure 2(b) shows the simulated spectrum for which the chemical shielding tensor orientation is determined by the local tetrahedral symmetry of the CH$_3$O unit. Here $\alpha = 180^\circ$, $\beta = 90^\circ$, $\gamma = 90^\circ$, the $z$ axis is oriented perpendicular to the O–C–O plane, the $x$ axis is located within the O–C–O plane and bisects the O–C–O angle. The $x$ axis is of course perpendicular to the $Y$ and $Z$ axes. Figure 5. Projection of the local CH$_3$O unit in polyoxymethylene showing the orientation of the chemical shift tensor in the molecule, obtained by transforming the axis system in Fig. 3 over the Euler angles 180°, 90°, 90°. The $Y$ axis points perpendicular to the plane of the drawing.

FIG. 5. Projection of the local CH$_3$O unit in polyoxymethylene showing the orientation of the chemical shift tensor in the molecule, obtained by transforming the axis system in Fig. 3 over the Euler angles 180°, 90°, 90°. The $Y$ axis points perpendicular to the plane of the drawing.

V. CONCLUSIONS

The dipolar correlated $^{13}$C chemical shift NMR technique can be used to determine rather accurately the orientation of a chemical shift tensor in a semicrystalline polymer like polyoxymethylene. The orientation is completely determined by the local symmetry.
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