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Synthetic Extracellular Matrices with Nonlinear Elasticity Regulate Cellular Organization

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Supporting Information

ABSTRACT: One of the promises of synthetic materials in cell culturing is that control over their molecular structures may ultimately be used to control their biological processes. Synthetic polymer hydrogels from polyisocyanides (PIC) are a new class of minimal synthetic biomaterials for three-dimensional cell culturing. The macromolecular lengths and densities of biofunctional groups that decorate the polymer can be readily manipulated while preserving the intrinsic nonlinear mechanics, a feature commonly displayed by fibrous biological networks. In this work, we propose the use of PIC gels as cell culture platforms with decoupled mechanical inputs and biological cues. For this purpose, different types of cells were encapsulated in PIC gels of tailored compositions that systematically vary in adhesive peptide (GRGDS) density, polymer length, and concentration; with the last two parameters controlling the gel mechanics. Both cancer and smooth muscle cells grew into multicellular spheroids with proliferation rates that depend on the adhesive GRGDS density, regardless of the polymer length, suggesting that for these cells, the biological input prevails over the mechanical cues. In contrast, human adipose-derived stem cells do not form spheroids but rather spread out. We find that the morphological changes strongly depend on the adhesive ligand density and the network mechanics; gels with the highest GRGDS densities and the strongest stiffening response to stress show the strongest spreading. Our results highlight the role of the nonlinear mechanics of the extracellular matrix and its synthetic mimics in the regulation of cell functions.

INTRODUCTION

An increasing number of reports reveal that the mechanical properties of the extracellular matrix (ECM) play a crucial role in the regulation of cell function.1,2 Adherent cells attach to the matrix via integrin–matrix protein binding and sense ECM mechanics with the aid of a large number of adhesion-associated proteins and mechanotransduction pathways.3 These physical signals synergize with the chemical signals and simultaneously and coherently orchestrate cell fate, including cellular organization, proliferation, migration, (stem cell) differentiation, and self-renewal.2,3 It should be noted that in vivo, cells reside in a soft three-dimensional (3D) microenvironment and it has become increasingly clear that (stiffer) 2D substrates are insufficiently capable to simulate the complex processes that occur in 3D and thus are not representative of a realistic situation. When 3D cell–matrix adhesion interactions are taken as an example, they differ from their 2D counterpart in the content of αvβ3 and αvβ5 integrins, paxillin, tyrosine phosphorylation of focal adhesion kinase (FAK), and other cytoskeleton proteins.4 Hence, to replicate the in vivo-like settings, the establishment of accurate 3D culture models has become crucial.4,5

The natural ECM is a sophisticated system that is composed of numerous elements that together provide the right chemical, biological, and mechanical environment for cells. From an architectural point of view, the fibrous components of the ECM play a main role in maintaining the structural integrity of the system and, as such, contribute to the bulk mechanical properties.6 As a result of their fibrous nature and high persistence lengths, these biopolymers are able to form stiff networks at very low concentrations, with large pores that allow for diffusion of large molecules as well as cell migration. In addition, these biopolymer networks typically possess intriguing mechanical properties: they become many times stiffer when a small strain is applied. This effect, which is known as stress-stiffening or strain-stiffening,7 enables natural tissues to dynamically adjust the local mechanics in response to small cellular forces, generated when cells anchor to the network and physically pull it. Stress-stiffening is believed to play a role in the prevention of tissue rupture, transduction of cellular forces,8 enabling of contractile cells to communicate...
Figure 1. Structure and hierarchical assembly of the polyisocyanide (PIC)-based gels as a reductionist model for the ECM. (a) The semiflexible network is composed of bundles of polymer chains with a broad pore size distribution ranging from nanometers to microns. The polymer contour length, \( L_C \), and the density of peptide adhesion sites, given by the average distance \( d \) between two adjacent groups, was varied in this work. (b) The molecular structure of the polymer is shown in red, the GRGDS peptide is shown in blue, and the linking group is shown in pink, matching with the colors in panel a. (c) The PIC polymers reversibly gel above the gelation temperature \( T_{gel} \sim 20 \, ^{\circ}C \).

Recently, we introduced polyisocyanide (PIC)-based hydrogels that are fully synthetic but strongly biomimetic in architecture and mechanics.\(^{21,22}\) On deformation beyond a critical stress, \( \sigma_c \), PIC gels can stiffen up a factor of 100, analogous to fibrin and collagen. The key mechanical parameters in PIC gels, i.e., \( \sigma_c \) and the stiffening index \( m \), metrics for the sensitivity and the responsiveness toward stress, respectively, indeed are similar to those found in these nature-derived gels. More importantly, in PIC gels, these parameters are readily controlled by the molecular structure, the concentration and external conditions, such as temperature and salt concentrations.\(^{22-26}\) Moreover, as a thermoresponsive polymer, PIC chains bundle up and form interconnected fibers above the gelation temperature (typically \( T_{gel} \sim 20 \, ^{\circ}C \)) and the gels disassemble back to polymer solutions below \( T_{gel} \), which enables researchers to easily encapsulate and harvest their cells. Earlier work has explored the use of PIC hydrogels as a guide for stem cell differentiation in 2D\(^{27}\) and 3D\(^{10}\) cultures, to support prevascularization, and as organotypic culturing support.\(^{28}\)

In this manuscript, we study how changes in the structure of the PIC polymer, i.e., the contour length \( L_C \) and the adhesive peptide density \( D \) on the polymer chain impact cellular behavior. For this study, we prepared a series of PIC polymers of different molecular weights and equipped them with adhesive peptides based on GRGDS (Figure 1a,b). We hypothesize that upon integrin-mediated binding of cells to the adhesive peptides on the polymer, the cells are able to generate internal stresses within the network and, as a result of the stress-sensitivity of PIC hydrogels, they are able to change the local mechanical microenvironment. By increasing/decreasing \( L_C \), we change the gel stiffness, but more importantly, the sensitivity to stress without changing its concentration.\(^{22}\) Through the peptide density, we tune the number of cell-polymer contacts and with that, the efficiency of stress transfer from the cell to its microenvironment. In our study, we first probed different cell lines. As the most stringent
effects were noticed with adipose-derived stem cells, we subsequently studied those more in-depth.

**EXPERIMENTAL SECTION**

**Synthesis of Polyisocyanides.** Polyisocyanides were synthesized as previously reported. In short, the isocyanide monomer was dissolved in freshly distilled toluene and stirred. Two parameters were adjusted: (i) the ratio of azide-functionalized and total monomer and (ii) the ratio Ni²⁺ to total monomer. All ratios are given in Table 1.

### Table 1. Polymers and Characterization

<table>
<thead>
<tr>
<th>Polymer</th>
<th>[Ni²⁺]/[M] ratio</th>
<th>Mₐ (kg mol⁻¹)</th>
<th>Lₐ (nm)</th>
<th>&lt;d&gt; (nm)</th>
</tr>
</thead>
<tbody>
<tr>
<td>PIC1</td>
<td>1:5000</td>
<td>562</td>
<td>223</td>
<td>10</td>
</tr>
<tr>
<td>PIC2</td>
<td>1:5000</td>
<td>757</td>
<td>228</td>
<td>10</td>
</tr>
<tr>
<td>PIC3</td>
<td>1:5000</td>
<td>558</td>
<td>222</td>
<td>25</td>
</tr>
<tr>
<td>PIC4</td>
<td>1:5000</td>
<td>539</td>
<td>214</td>
<td>50</td>
</tr>
<tr>
<td>PIC5</td>
<td>1:5000</td>
<td>568</td>
<td>225</td>
<td>100</td>
</tr>
<tr>
<td>PIC6</td>
<td>1:5000</td>
<td>568</td>
<td>225</td>
<td>3</td>
</tr>
<tr>
<td>PIC7</td>
<td>1:1000</td>
<td>263</td>
<td>106</td>
<td>10</td>
</tr>
<tr>
<td>PIC2</td>
<td>1:5000</td>
<td>575</td>
<td>228</td>
<td>10</td>
</tr>
<tr>
<td>PIC8</td>
<td>1:7000</td>
<td>589</td>
<td>236</td>
<td>10</td>
</tr>
</tbody>
</table>

The appropriate amounts of monomers and catalyst solution Ni(ClO₄)₂·6H₂O (0.1 mg ml⁻¹) were dissolved into toluene/absolute ethanol (9:1) were dissolved into toluene and the HNCO isocyanide concentration was adjusted to 50 mg mL⁻¹. The mixture was stirred at room temperature, and the progress of the reaction was followed by IR-ATR (disappearance of the characteristic isocyanide absorption at 2140 cm⁻¹). Once the polymerization was complete, the polymer was precipitated into diisopropyl ether under vigorous stirring and collected by centrifugation. The polymer was dissolved in dichloromethane, precipitated for another two rounds, and air-dried to yield the polymer. No GRGDS present in PIC6.

The appropriate amounts of monomers and catalyst solution Ni(ClO₄)₂·6H₂O (0.1 mg ml⁻¹) were dissolved into toluene/absolute ethanol (9:1) were dissolved into toluene and the HNCO isocyanide concentration was adjusted to 50 mg mL⁻¹. The mixture was stirred at room temperature, and the progress of the reaction was followed by IR-ATR (disappearance of the characteristic isocyanide absorption at 2140 cm⁻¹). Once the polymerization was complete, the polymer was precipitated into diisopropyl ether under vigorous stirring and collected by centrifugation. The polymer was dissolved in dichloromethane, precipitated for another two rounds, and air-dried to yield the polymers as off-white solids. The molecular weight of the polymer was determined by viscometry (dilute solutions in acetonitrile) using the empirical Mark–Houwink equation \( \eta = K M^a \), where \( \eta \) is the experimentally determined intrinsic viscosity, \( M_v \) is the viscosity-determined molecular weight, and the Mark–Houwink constants \( K \) and \( a \) depend on polymer characteristics and solvent, temperature, etc. We use parameters previously determined²⁰ for other polyanions: \( K = 1.4 \times 10^{-9} \) and \( a = 1.75 \). The results are given in Table 1.

**Conjugation of Adhesive Peptides.** The peptide GRGDS was first conjugated to the BCN-containing spacer following earlier described protocols.¹⁰ The peptide (H-Gly-Arg-Gly-Asp-Ser-OH, Bachem, Germany) was dissolved in borate buffer (pH = 8.4) at 6 mg mL⁻¹. The dissolved BCN-NHS linker (Synapse, The Netherlands) in DMSO (6 mg mL⁻¹) was added to the peptide solution in borate buffer in a 1:1 molar ratio and stirred on the stirring plate for 24 h at room temperature. The formation of the BCN−GRGDS conjugate was confirmed by mass spectrometry (\( M = 911.4 \text{ g mol}^{-1} \)). The azide-appended polymer (PIC1−PIC5, PIC7, PIC8) was dissolved in acetonitrile (2.5 mg mL⁻¹), and the appropriate volume of the BCN−GRGDS solution in borate buffer (95% molar equivalent of the corresponding azide amount on the polymer) was added. The solution was stirred for 24 h at room temperature. The polymer−peptide conjugates were precipitated in diisopropyl ether, collected by centrifugation, and air-dried for 24 h.

**Rheological Analysis of Polymer Hydrogels.** For the mechanical analysis of the gels, a stress-controlled rheometer (Discovery HR-1 or HR-2, TA Instruments) with an aluminum or steel parallel plate geometry was used (diameter = 40 mm, gap = 500 µm). All samples were loaded onto the rheometer plate in the liquid state at \( T = 5 ^\circ C \) followed by a temperature ramp to \( T = 37 ^\circ C \) at a rate of 1.0 °C min⁻¹. The moduli were measured in the linear viscoelastic regime at amplitude of \( \gamma = 0.02 \) or 0.04 and a frequency of \( \omega = 1.0 \text{ Hz} \). The sample was allowed to equilibrate at 37 °C prior to the nonlinear measurements. Here, the gel was subjected to a constant presress of \( \sigma_0 = 0.5 \) to 200 Pa, and the differential modulus \( K' \) was probed with a small superposed oscillatory stress at frequencies of \( \omega = 10 \) to 0.1 Hz (reported data at \( \omega = 1 \text{ Hz} \)). The oscillatory stress was at least 10 times smaller than the applied presress.

**Cell Culture and Encapsulation.** Human bladder smooth muscle cells were purchased from ScienCell and cultured in smooth muscle cell medium with growth supplement (ScienCell, USA). T24 and Hela cells were purchased from ATCC; SKRC52 cells were obtained from a mediastinal metastasis of a primary RCC2. All cancer cells were cultured in RPMI 1640 with GlutaMax (Gibco, Thermo Fisher, USA). Human adipose-derived stem cells were obtained from the Radboud Biobank and cultured in minimum essential medium eagle (α-MEM) (Invitrogen, Thermo Fisher, USA). All media were supplemented with 10% fetal bovine serum (Sigma-Aldrich, USA) and 1% penicillin/streptomycin (final concentration of 100 IU/mL penicillin and 100 μg/mL streptomycin, Gibco, Thermo Fisher, USA).

Dry PIC polymers were sterilized by UV for 20 min and then dissolved in medium for 24 h at 4 °C. Cells were harvested by trypsin treatment once they reached 100% confluence and were resuspended in fresh medium. Cell densities were determined by a LUNA-FL dual fluorescence cell counter. Cells were mixed with the polymer solution on ice with a predetermined ratio to achieve the required cell density and polymer concentration. After mixing, the solutions were transferred to 48-well plates (Corning, USA) or 8-well chambered cover slides (Sigma-Aldrich, USA) and heated to 37 °C where gelation occurred. After gel formation, culture medium (at 37 °C) was added onto the samples. Then all samples were subject to standard cell culture conditions (37 °C, 5% CO₂). Note that no stem cell differentiation kit was added into the culture medium of the hACs, and the cells are expected to maintain their stemness.¹⁵

**Bright Field Imaging and Cell Morphology Analysis.** Bright field images of cells encapsulated in hydrogels were acquired on a Leica DC200 microscope. Outlines of representative hACs were plotted manually, and the quantitative analysis of circularity was performed by Fiji. A paired sample t test was used to determine the statistical significance.

**Cytoskeleton Staining and Confocal Microscope Imaging.** Gels with encapsulated cells were washed with 0.9% NaCl and then fixed with 3% paraformaldehyde in 0.9% NaCl for 40 min. After fixation, the samples were permeabilized with 0.1% Triton X-100 in 0.9% NaCl for 10 min and blocked with 1% BSA in 0.9% NaCl for 30 min. They were then incubated with Texas Red-X Phalloidin (50 IU/mL in 1% BSA, 0.9% NaCl, Sigma-Aldrich, USA) for 1 h and DAPI (5 mg mL⁻¹ in 0.9% NaCl, Thermo Fisher, USA) for 10 min. All procedures above were performed at 37 °C. An Olympus FluoView 1000 confocal laser scanning microscope was used for fluorescence imaging. The temperature of the sample was kept at 37 °C by the heating element of the microscope.

**Cell Proliferation Assay.** (a) WST-1: Culture medium was gently removed, the new medium supplemented with the cell proliferation reagent WST-1 (Roche, Switzerland) at a final concentration of 1:10 (WST-1 stock solution/total working solution) was added, and the culture plates were incubated at 37 °C, 5% CO₂ for 2 h. The absorbance was measured at \( \lambda = 450 \text{ nm} \) with a plate reader (PerkinElmer 1420 Multilabel Counter). All samples were measured in triplicates. (b) Quant-IT PicoGreen dsDNA Assay Kit: The amount of dsDNA in each sample was quantified according to the kit manual (Invitrogen, Thermo Fisher, USA). In brief, 28.7 μL of sample, 71.3 μL of 1X PicoGreen solution, and 100 μL 1X TE were mixed and incubated in a 96-well plate in the dark for 10 min and the fluorescence was read with a plate reader (λ_{exitation} = 485 nm, λ_{emission} = 528 nm). All samples were measured in triplicates.
RESULTS AND DISCUSSION

Synthesis of the PIC-GRGDS Polymers and Gel Preparation. The generic polymer structure of the materials used in this work is shown in Figure 1b. For precursors, we prepared azide-appended PIC polymers PIC1–PIC8 by copolymerization of the N3-functionalized monomer with the nonfunctional monomer, following earlier reported procedures.10 The polymer contour length, \( L_c \), was tuned by changing the total monomer to initiator ratio and calculated from the viscosity-averaged molecular weight.23 Although at gel formation, the polymers bundle and form infinitely long networks, the length of the individual polymer chains is an important parameter in the linear and nonlinear mechanical properties of the gels.22 After purification of the polymers, the appropriate amounts of the GRGDS peptide, previously equipped with a BCN group,10 were added to the polymers, and the GRGDS-decorated polymers were collected after precipitation. As their concentration is very low, a quantitative characterization of the degree of peptide conjugation is challenging. On the basis of the high reaction rate of this conjugation chemistry31 and the results of earlier reports,22 we assume complete conversion and we present GRGDS spacings, \( d \), based on the azide monomer fractions (Table 1, series PIC7–PIC8). Although the experimental values of the viscosity-averaged molecular weights, \( M_w \), of PIC2 and PIC8 are very similar, we do find the expected differences in the mechanical properties of the corresponding gels.

For analysis and cell culture studies, the appropriate amount of the azide or GRGDS-functionalized polymers were dissolved from the viscosity averaged molecular weight.22 Although at gel formation, the polymers bundle and form infinitely long networks, the length of the individual polymer chains is an important parameter in the linear and nonlinear mechanical properties of the gels.22 After purification of the polymers, the appropriate amounts of the GRGDS peptide, previously equipped with a BCN group,10 were added to the polymers, and the GRGDS-decorated polymers were collected after precipitation. As their concentration is very low, a quantitative characterization of the degree of peptide conjugation is challenging. On the basis of the high reaction rate of this conjugation chemistry31 and the results of earlier reports,22 we assume complete conversion and we present GRGDS spacings, \( d \), based on the azide monomer fractions (Table 1, series PIC7–PIC8). Although the experimental values of the viscosity-averaged molecular weights, \( M_w \), of PIC2 and PIC8 are very similar, we do find the expected differences in the mechanical properties of the corresponding gels.

For analysis and cell culture studies, the appropriate amount of the azide or GRGDS-functionalized polymers were dissolved in \( \alpha \)-MEM medium at 4 °C for 24 h. Instantaneous and reversible gelation takes place upon heating above \( T_{gel} \) (Figure 1c).

Mechanical Characterization. The PIC hydrogels were subjected to rheological analysis (Table 2). At low strain, we measure the storage modulus \( G' \) in the linear viscoelastic regime as a function of temperature (Figure S1). At high stresses, the gels reversibly stiffen and we describe the mechanical properties by the more relevant differential modulus \( K' = \partial G'/\partial \gamma \), where \( \sigma \) and \( \gamma \) are the stress and strain, respectively. Note that below a critical stress \( \sigma_c \), i.e., at \( \sigma < \sigma_c \), \( K' = G' \), and at \( \sigma > \sigma_c \), \( K' \) depends on the applied stress \( G' \); \( G' \sim \sigma^m \), where \( m \) is the stiffening index. The parameters \( \sigma_c \) and \( m \) represent the sensitivity and responsiveness to macroscopic deformation.

Polymers PIC1–PIC6 have the same contour length \( L_c \), but different GRGDS loading (\( \delta \)), and all form soft hydrogels with \( G' = 230 \) to 590 Pa at polymer concentration \( \varepsilon = 2 \) mg mL\(^{-1} \) in medium at a physiological temperature of \( T = 37 \) °C (Figure 2a). Beyond the linear viscoelastic regime, the gels stiffen. Again at 2 mg mL\(^{-1} \), we find for P1–P6, a critical stress \( \sigma_c = 19 \) to 35 Pa and a stiffening index \( m \approx 1 \). As expected, for polymers with the same molecular weight, the curves collapse to a single master curve when the data is scaled to the storage modulus in the linear viscoelastic regime \( G' \) and \( \sigma_c \) (Figure S1). We do not find a clear correlation between the mechanical properties of the gel and the peptide loading (Table 2). Low peptide densities stiffen the gel slightly, but at the highest GRGDS densities, the gel softens. This nonlinear behavior results from a balance in effects introduced with the peptide. On one hand, the peptide makes the polymer slightly more hydrophilic, which increases \( T_{gel} \) and thus decreases \( G' \) at \( T = 37 \) °C. On the other hand, the peptides add physical interactions between polymer chains, which will increase \( G' \). Overall, however, all gels are relatively soft, i.e., much softer than many other (synthetic) gels used for cell culture experiments, and display stress sensitivities that are biologically accessible.23 Gels S1–S6 of polymers with the same peptide density \( D \) but of different lengths show an increase in both \( G' \) and \( \sigma_c \) with increasing \( L_c \) (Figure 2b). We used the gels at two different concentrations: \( \varepsilon = 1.5 \) and 3 mg mL\(^{-1} \), which gives gels with \( G' = 100 \) to 1000 Pa, \( \sigma_c = 14 \) to 41 Pa, and \( D = 41 \) or 83 \( \mu \)M. The adhesive peptide concentrations in our gels are typically about an order of magnitude lower than in other commonly used synthetic gels.23

We find that softer gels are more stress-responsive (Figure 2c), that is, they stiffen at lower stresses (low \( \sigma_c \)). Interestingly, the stiffening index of gel series S1–S6 negatively correlates to the critical stress (Figure 2d), which underlines that softer PIC hydrogels with a higher stress-sensitivity also possess a higher responsiveness (high \( m \)). The fibrous architecture of the gels does not change significantly with the polymer chain length (in this studied regime)24,36 which means that only a change in the architecture of the network is expected from the change in concentration, which for a concentration doubling gives a minor reduction of the mesh size \( \xi \) as follows: \( \xi \sim c^{-0.5} \).

Adhesive Peptide Density: Cell Proliferation vs Spreading. For cell culture studies, the cells were suspended in a cold (0–5 °C) polymer solution in medium and immediately warmed to 37 °C to form 3D cell–gel constructs. Then the samples were cultured at standard conditions for 7 days. Bright field images show that the cell morphologies depend on the cell type and GRGDS concentration but much less on the gel properties (Figure 3a,b). Human bladder smooth muscle cells (hSMCs), HeLa, SKRC52, and T24 cells tend to proliferate and organize into multicellular spheroids with diameters of tens of microns in all polymers (see Figure 3a and Figures S2–S5 for a full overview). The spatial distribution of cell nuclei and the arrangement of F-actin, however, differs for each cell type (Figure 3c).
cell functions. Although commonly used in 2D cell studies, concentrations of the synthetic biomimetic hydrogels seem most sensitive to the RGD peptide density. Cell protrusion and elongation are seen earlier in spheroids but rather show distinct morphological changes with stem cells (hASCs) do not organize into large multicellular structures is a very complex mechanism with rich biological content, where besides cell–matrix interactions, cell–cell contacts play a major role. In this work, however, we will not discuss this in further detail but restrict ourselves to the stem cells that seem most sensitive to the RGD concentrations of the synthetic biomimetic hydrogels.

We studied cell spreading as an indicator for a plethora of cell interactions in 2D planar surfaces. In this work, we focus on cell spreading on planar surfaces proved the existence of a critical adhesive ligand density (e.g., 70 nm for mesenchymal stem cells on nanopatterned PEG matrix) for efficient cell spreading and focal adhesion formation. 3D cell–matrix adhesions, however, are distinct from focal adhesions that are found on two-dimensional planar surfaces. In this work, we focus on cell spreading in a 3D fibrous network with a controlled ligand density.

When polymer length and polymer concentration are fixed, our results reveal that microenvironments rich in adhesion sites promote stem cell spreading and the formation of cellular networks by providing more molecular anchoring points for integrins on the cell membrane. In other words, it is easier for cells to anchor to the artificial ECM and initiate actin polymerization, myosin contraction, and adhesion protein recruitment. In our case, the minimum required adhesive peptide density for hASCs to spread is ∼55 μM (Figure 3b). This density corresponds to a ∼10 nm distance between adjacent peptides on a polymer chain, a distance that is of the same order of magnitude as the length of the α5β1 integrin head (that is overexpressed in human mesenchymal stem cells). When two adjacent peptides are too far apart, hASCs seem unable to generate enough traction to modulate their morphology. Once more, we iterate that the RGD density used in our study is low compared to the RGD content inside other synthetic cell-laden hydrogels. We propose that the bundling of PIC polymers results in the clustering of RGD peptides, which results in more effective cell spreading. Notably, we also encapsulated hASCs in PIC1–PIC6 with a higher proliferation rates of these cell types are positively correlated to the density of adhesive ligands (Figure 3d–g): gels with the highest GRGDS loading show highest proliferation rates. The relatively small difference in the mechanical properties in PIC1–PIC6 has a minor effect on cell behavior. Human adipose-derived stem cells (hASCs) do not organize into large multicellular structures, while the density of adhesive ligands (Figure 3d) and between the stiffening index $m$ and $\sigma_c$ for PIC1–PIC6; 1.5 mg mL$^{-1}$ (solid symbols) and 3.0 mg mL$^{-1}$ (open symbols). The dashed line is a guide to the eye. The last two diagrams clearly show that the more stress-sensitive gels are both softer and more responsive toward applied stress. All measurements were done in α-MEM medium at 37 °C.
cell density (200000 cells mL\(^{-1}\)) and found a similar trend in morphology and proliferation (Figures S8 and S9).

Nonlinear Mechanics of the ECM Regulates the Organization of hASCs. Nonlinear mechanics is frequently observed in gels of natural polymers, but to this date, many biological effects of stress-stiffening and their mechanisms still remain to be unraveled.\(^{1,10}\) We prepared six gels S1–S6 with different critical stresses and thus different responses to stress that is applied externally (rheometer) or internally (cells) to screen the influence of ECM stress-stiffening on cell spreading of 3D hASC cultures. The PIC polymers were conjugated with a fixed density of adhesive peptides with \(d = 10\) nm to provide the essential amount of attachment sites. After culturing for 3 days, the cells encapsulated in S1–S6 were analyzed (Figure 4) using bright field microscopy (Figure 4a) and after staining the nuclei and F-actin, the cells were analyzed using confocal microscopy (Figure 4b). The extent of spreading of the cells was quantified and measuring the circularity of ten representative cells (Figure 4c,d). At low gel concentrations (\(c = 1.5\) mg mL\(^{-1}\)), the stem cells show significantly better spreading (smaller circularity) in samples with a smaller critical stress (series S1–S3). At 3 mg mL\(^{-1}\) (S4–S6), a plateau is reached (Figure 4e). We did not detect any significant proliferation until day 3, which is in line with the results from P1–P6 (Figure 4f).

A wide variety of cellular processes, including sensing of ECM mechanics, cell–cell communication, on/off switching of functional binding sites, and alteration of enzymatic ECM degradation,\(^{11}\) involve cell anchoring to the ECM fibers and the straining of the fibers that induces a stiffening response in the matrix. The storage modulus (i.e., the stiffness in the linear viscoelastic regime) of the PIC gels falls into the category of soft biomaterials,\(^{18}\) which means that primarily the difference in critical stress governs the real mechanical properties of the microenvironment of the encapsulated cells. A lower critical stress implies that the material is more sensitive to cell-induced matrix stiffening. For gels that display large stiffening indices, i.e., gels that are more responsive toward contractile stresses applied by the cell, this effect is further enhanced. In these fibrous gels, cells are able to adhere and pull the PIC fibers, stiffen them, and accumulate enough traction force for the observed morphological changes. The culturing in gels that have a higher \(\sigma_c\) and a lower \(m\) leads to the opposite effect. These materials remain relatively static; i.e., the cells are able to anchor due to the sufficient amount of adhesive peptides present but are unable to generate sufficient traction force to spread. Therefore, cells tend to maintain the rounded morphology and do not spread out. Despite the difference in initial low-stress plateau modulus, the role of nonlinear mechanics at higher stress is more dominant as these soft gels can stiffen up to kilopascals with external stress (Figure 2b).

In addition to the effects originating from changes in the macroscopic (nonlinear) mechanical properties, we observed cell alignment in some experiments. Despite the low cell density in the samples (down to 20000 hASCs mL\(^{-1}\)), we found that in some of our gels, cells aligned and protrusions form in a common direction (Figure 4a panel S1). The alignment implicates that isolated cells are able to induce long-range interactions by matrix remodeling and/or that fiber alignment occurs in the semiflexible polymer network, which allows neighboring cells to sense the fiber orientation and
align. Similar findings have been observed and discussed earlier by several research groups.\textsuperscript{8,44} These observations are in agreement with cell studies in collagen and fibrin. We note that fiber alignment will enhance (local) stress-stiffening behavior in PIC gels\textsuperscript{45} and subsequently in influence cellular behavior. In addition, we point out that, over time, stem cells will secrete natural ECM that will contribute to the mechanical micro-environment of the cell.

\section*{CONCLUSIONS}

It is universally acknowledged that the storage modulus is a key component of hydrogel mechanics that affects cellular behavior in 3D cell cultures. More recently, stress relaxation was identified as another important contributor. We wish to add to this perspective that nonlinear mechanics also strongly contribute to the mechanical spectra that cells are able to sense. In this work, we provide insight into the correlations between cellular biophysics and stress-stiffening of biomimetic polymer gels. In the presence of cells that apply contraction forces to the gel, the stiffness no longer remains a “static” constant but becomes a dynamic parameter that responds to cell–gel interactions, which can be tuned by the design of the polymer. In biological or biomimetic, fibrillar, and strain-stiffening matrices, it is insufficient to describe the mechanical properties of the gel by its “static” stiffness alone.

For cell culturing in synthetic matrices, which offer the advantage of tunable mechanical properties, one needs to engineer cell attachment sites. The density and distribution of adhesive peptides will play yet another important role in the accomplishment of desired cell behavior. Finally, different cells, cultured in the same matrix, will also respond differently. Our results once more underline that all these factors should be included in a comprehensive experimental design, but at the same time, it is still difficult to predict which of the parameters will dominate behavior and therefore be the most urgent to optimize.

Our minimal PIC hydrogel model is particularly suitable as a guide to further design soft biomimetic materials. As the next step of research, mechanical analyses of both bulk tissue constructs and the local niche is needed, for instance, it would be of great interest to characterize the (variation in) local matrix stiffness using microrheology. Ultimately, the comprehensive understanding of interactions at the cell–matrix interface should be utilized to benefit regenerative medicine and tissue engineering.
Diagrams S1–S9, showing the mechanical properties of gels of P1–P6 (Figure S1); bright field cell culture images at day 1, 3, and 7 in gels of P1–P6 (2.0 mg mL\(^{-1}\)) with hBSMCs (Figure S2); HeLas (Figure S3); T24s (Figure S4); SKRCS2s (Figure S5) and hASCs (Figure S6); bright field cell culture images at day 0, 1, and 2 of hASCs in gels S1–S6 at \(c = 1.5\) and 3.0 mg mL\(^{-1}\) (Figure S7); bright field cell culture images at day 1, 3, and 7 of hASCs in gels of P1–P6 at \(c = 2.0\) mg mL\(^{-1}\) at higher cell density 20000 cells mL\(^{-1}\) (Figure S8); crude WST-essay results (Figure S9); LIVE/DEAD staining of hASCs at 48h in P2 (Figure S10) (PDF).

**REFERENCES**


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